In situ coupling of N-doped lignin-derived carbon-encapsulated CoFe-CoxN heterojunction for oxygen evolution reaction

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In situ coupling of N-doped lignin-derived carbon-encapsulated CoFe-Co $_x$ N heterojunction for oxygen evolution reaction

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Abstract

Exploring highly active and stable electrocatalyst for oxygen evolution reaction is important for the development of water splitting and rechargeable metal-air batteries. Herein, a hybrid electrocatalyst of CoFe alloy and Co_xN heterojunction encapsulated and embedded in N-doped carbon support (CoFe-Co_xN@NC) was in situ coupling via a pyrolysis process of a novel coordination polymer from lignin biomacromolecule. CoFe-Co_xN@NC exhibited an excellent OER activity with a low overpotential of 270 mV at 10 mA·cm⁻² and stability with increment of 20 mV, comparable to commercial Ir/C. DFT calculations revealed that Co_xN and N-doped grapheme encapsulation can reduce the binding strength between *O and CoFe alloy, prevent

metals leaching and agglomeration, and improve electron transfer efficiency, thereby, remarkably enhancing the OER activity and stability. In situ coupling strategy of alloy and nitride heterojunction on N-doped lignin-derived carbon provided a promising and universal catalyst design for the development of renewable energy conversion technologies.

Keywords : Lignin-derived carbon; CoFe alloy; Heterojunction; Oxygen evolution reaction; Electrocatalysis.

1. Introduction

Electrochemical water splitting powered by solar, wind and tidal renewable energies is an ideal green and costeffective hydrogen production path, and there are no carbon-containing compounds discharged^{1,2}. Water splitting consists of two half reactions: hydrogen evolution reaction (HER) on the cathode and oxygen evolution reaction (OER) on the anode. The OER process suffers from multiple steps of proton-coupled electron transfer, thereby causing very sluggish reaction kinetics and high overpotential ^{3,4}. Up to now, noble metals and metal oxides, such as Ir and Ru, are the state-of-the-art catalysts for the OER process, however, their scarcity, high costs and poor stability has severely hindered their widespread commercialization ⁵⁻⁷. Therefore, it is highly desirable to search and develop highly efficient, stable, and cost-effective OER electrocatalysts ^{3,8}.

Owing to the intrinsic electron structure, special orbitals and high carrier mobility, transition metals Co-based and Ni-based catalysts have been widely investigated toward OER and perform great promise for potential applications $^{9-11}$. A wide variety of Co-based catalysts have been well investigated for OER, including metal oxides 12,13 , layered double hydroxides 14,15 , bimetallic alloys $^{16-20}$, metallic nitrides 21,22 and phosphides 23 . Co-based oxides and layered double hydroxides are limited toward electrocatalytic performance for their poor electrical conductivity, which hinder the electron transfer 11,24 . Doping Fe element in Co species adjusts the local electronic structure and enhances the active sites, which can boost the OER electrocatalytic activity $^{18,25-29}$. In addition, bimetallic alloys and Co_xN nitrides exhibit high OER catalytic activities, attributed to their superior electrical conductivity 22,30,31 . However, Co-based monometallic or bimetallic catalysts are usually unstable and oxidized during the OER process in alkaline solution 32 . What's more, bimetallic alloys show the strong binding with oxygen intermediates toward OER, that require the high overpotential to overcome the reaction barriers 25 .

Carbon-based nanomaterials have shown great potential in the design of metal based electrocatalysts for their cost-efficient, high specific surface area, excellent conductivity, and favorable durability in a harsh environment $^{27,28,33-37}$. However, the inert surface of carbon materials usually leads to the weak interaction between the metal nanoparticles and supports, which make it difficult to effectively regulate the electronic structure of metal nanoparticles and improve their catalytic activity 27,28 . Biorefinery lignin is the most promising carbon precursor instead of petroleum and coal resources, owing to its second largest natural organic polymer after cellulose 38,39 . In addition, with a three-dimensional network structure formed by phenylpropane units, lignin is rich in phenolic hydroxyl groups, alcoholic hydroxyl groups and part of carboxyl groups, and its carbon content is as high as $60\%^{40,41}$. Moreover, lignin-derived carbon have been reported as the catalyst support in the applications of electrocatalysis^{42,43}, photocatalysis degradation^{44,45}, Fischer-Tropsch synthesis^{46,47} and other fields ^{48,49} in the last few years, but most of the lignin-derived carbon catalyst are prepared using direct carbonization of lignin followed by impregnation of metal active ingredients, of which was easily dissoluble and difficult to effectively inhibit their agglomeration, causing the inferior activity and durability ⁵⁰.

In this work, making full use of the inherent properties and abundant carboxyl/hydroxyl functional groups of lignin biomacromolecule to precisely coordinate with transition metal ions with Co^{2+} and Fe^{3+} by aqueous self-assembly process, and form the lignin-based CoFe bimetallic supramolecules, then co-doped with nitrogen precursor urea in situ pyrolyzing at high temperature to obtain the desired CoFe biometallic active nano

particles. The effects of the pyrolysis temperature, urea dosage, Fe/Co metal ratio and metal/lignin ratio during synthesis process on the structure and OER electrocatalytic activity were investigated. To our delight, Co_xN component was in situ generated other than CoFe alloy during the pyrolysis and doping of urea revealed by the systematic characterization, and coupled with CoFe alloy to form the CoFe-Co_xN heterojunction encapsulated by the graphic carbon layers and located on the nitrogen-doped lignin-derived shell-core carbon support (CoFe-Co_xN@NC), that optimize the adsorption/desorption strength between the OER intermediates and metallic active site, disperse and decrease the size of CoFe-Co_xN heterojunction, and prevent their aggregation, endowing the as-synthesized catalyst CoFe-Co_xN@NC exhibited excellent OER electrocatalytic activity and stability in an alkaline medium.

2. Experimental section

2.1 Materials

Enzymatic hydrolysis lignin (EHL) isolated from the corncob bio-refinery residue through alkali dissolving, acid precipitation, water washing, drying, ball milling and sieving, was provided by Shandong Longli Biotechnology Co., Ltd. Iron chloride hexahydrate (FeCl₃*6H₂O, AR 99%), cobalt nitrate hexahydrate (Co(NO₃)₂*6H₂O, AR 99%), urea (CO(NH₂)₂, AR 99%), ammonia (NH₃*H₂O, AR 25-28%), ethanol (C₂H₅OH, AR 99.7%), commercial iridium carbon (20 wt%) were purchased from Shanghai Aladdin Chemical Co., Ltd., monochloro acetic acid (C₂H₃ClO₂, AR 98%) was purchased from Shanghai Macklin Biochemical Technology Co., Ltd., and acetone (C₃H₆O, AR 98%), hydrochloric acid (HCl, 36.0-37.0%), nitric acid (HNO₃, 65.0-68.0%), sodium hydroxide (NaOH, AR 96%), potassium hydroxide (KOH, AR 85%) were produced by Guangzhou Chemical Reagent Factory. Nafion solution (D520, 5 wt%) was supplied from DuPont. Carbon papers (TGP-H-060, 20 cmx20 cm, 0.19 mm) were produced by Toray Corporation (Japan).

2.2 Preparation of lignin-based CoFe supramolecules

Carboxylated enzymatic hydrolyzed lignin (EHL-COOH) was prepared by the aqueous method according to our previous report ⁵¹. Then 1.0 g EHL-COOH was dissolved in 100 mL pure water, while FeCl₃*6H₂O and Co(NO₃)₂*6H₂O ($n_{Fe}/n_{Co} = 8:0, 7:1, 6:2, 4:4, 2:6, 1:7, 0:8$) were added into 50 mL pure water under vigorous stirring. The mixture was added drop by drop into the aqueous solution of EHL-COOH and stirred for 10 min. The pH of above solution was adjusted to 8 by NH₃*H₂O and HCl. After aging for 12 h, the precipitate was centrifuged, washed with ultra-pure water, dried at 80 for 24 h, and ground to obtain the lignin-based CoFe supramolecules (EHL-COOH-CoFe).

2.3 Preparation of lignin-derived carbon CoFe-Co_xN@NC

Firstly, 1.0 g of EHL-COOH-CoFe was ground and mixed with a certain amount of urea, then was heated at a rate of 5 $*\min^{-1}$ and pyrolyzed at high temperature for 2 h under the argon atmosphere. The products were washed with 1 mol*L⁻¹ HCl and ultra-pure water, respectively, and dried at 80 for 12 h to obtain the bimetal-based functionalized nitrogen-doped lignin-derived carbon materials, denoted as CoFe-Co_xN@NC. The effects of metal ratio, pyrolysis temperature, the ratio of metal to lignin, and nitrogen content on the structure and electrocatalytic performance of CoFe-Co_xN@NC were systematically investigated. 1.0 g of EHL-COOH was mixed and ground with 1.0 g of urea, then pyrolyzed at 700 to obtain the metal free nitrogen-doped lignin-derived carbon materials (NC) as comparative sample.

2.4 Structural characterization of CoFe-Co_xN@NC

The crystal structure of the electrocatalyst was characterized by Rigaku Miniflex-600 XRD instrument, which used a Cu-K α radiated X-ray source with a scanning range of $2\theta = 5-80$ and a scanning rate of 10 °·min⁻¹.

The morphology and elemental composition were characterized by American FEI Tecnai G^2 F20 TEM, and its operating voltage was 200 kV. The degree of order and graphitization were characterized by LabRAM HR Evolution Raman spectrometer, with an excitation wavelength of 532 nm. Nitrogen adsorption/desorption isotherms were measured at 77 K with ASAP 2460 surface area and porosity analyzer. The elemental composition and valence state were characterized by Thermo Scientific K-Alpha + X-ray photoelectron spectrometer (XPS). All the peaks were calibrated according to the standard position of C 1s peak (284.8 eV). The percentage of doped metal in the sample was calculated by Varian 720 inductively coupled plasma OES spectrometer (ICP-OES). The band gap width of carbon materials was determined by PE Lambda 950 UV-Vis Diffuse Reflectance Spectrometer. The valence band and conduction band of carbon materials were tested by UV photoelectron spectrometer (Thermo Fischer Escalab 250 Xi).

2.5 Electrochemical measurements

All electrochemical measurements were performed in a three-electrode system using the Gamry Interface 1010 electrochemical workstation. Graphite rod and Hg/ HgO were used as the counter electrode and reference electrode, respectively. The working electrode was prepared as follows, 4 mg of carbon powder was added into 200 μ L of 0.25% nafion-ethanol solution. After ultrasonic dispersion of the powder, 50 μ L of the slurry was dropped onto the treated carbon paper, and then the carbon paper was held by electrode clamp as the working electrode. The loading capacity of the catalyst was 4 mg·cm⁻².

The electrochemical performances were investigated by using linear sweep voltammetry (LSV) and cyclic voltammetry (CV) techniques at room temperature. $1 \text{ mol}^*\text{L}^{-1}$ KOH solution was used as electrolyte, and the scanning rate recorded by all the polarization curves was $1 \text{ mV}^*\text{s}^{-1}$. Before LSV testing, the working electrode with loading catalyst was firstly activated by several CV cycles. All the measured data of the polarization curve were corrected by IR, and all the potentials were calculated according to the reversible hydrogen electrode (RHE). The expression is as follows:

E (vs. RHE) = E (vs. Hg/HgO) + 0.098 + 0.0591 × pH.

2.6 Density functional theory (DFT) calculations

DFT calculations were carried out using the Vienna ab Initio Simulation Package (VASP) ⁵². The Perdew-Burke-Ernzerhof (PBE) model in the form of the generalized gradient approximation (GGA) was used to describe the exchange-correlation potential. For all optimization calculations, the energy and force convergence criteria were set to 10^{-5} eV and $0.05 \text{ ev}^*\text{A}^{-1}$, respectively. Total energy was calculated using the Projector Augmented-Wave (PAW) method with a plane-wave cutoff energy of 400 eV. A gamma-centered K-point of 3x3x1 was chosen to describe the Brillouin zone.

3. Results and discussion

3.1 Synthesis and characterization of CoFe-Co_xN@NC

The CoFe-Co_xN@NC catalyst was synthesized by self-assembly and pyrolysis process, as shown in Fig. 1a. In brief, biorefinery lignin was carboxylated to improve its solubility and coordinated with Co^{2+} and Fe^{3+} to form the lignin-based CoFe supramolecules, followed by co-doping with urea and in situ pyrolyzing at high temperature to obtain the CoFe-Co_xN heterojunction on the nitrogen-doped lignin-derived carbon.

The crystal phase structure and graphitization degree of CoFe-Co_xN@NC samples were shown in Fig. 1 and Fig. S1. From the XRD patterns of the metal free nitrogen-doped lignin-derived carbon materials (NC) and CoFe-Co_{5.47}N@NC, it could be observed that the characteristic peaks of NC at the position of 24deg and 43deg, respectively, which belonged to the (002) and (100) crystal planes of graphitic carbon⁵³. When the

Fe/Co ratio gradually decreased from 7:1 to 1:7, the XRD peaks of (110) and (200) planes of Fe (PDF#06-0696) moved to a smaller angle and disappeared until the (111) and (200) planes of Co (PDF#15-0806) appeared, and all the samples had a broad peak around 26deg attributed to the (002) plane of graphitic carbon, as shown in Fig. 1b. The dominant phase is bimetallic CoFe alloy with diffraction peaks centered at 44.9deg and 65.3deg, which can be ascribed to the (110) and (200) crystal planes of CoFe alloy (PDF#49-1568). The diffraction peaks of 43.7deg, 50.8deg and 74.9deg corresponding to the (111), (200) and (220) crystal planes of Co_{5.47}N (PDF#41-0943) were also observed ⁵⁴. The XRD peak of CoFe-Co_{5.47}N@NC moved towards a higher angle (~ 0.4deg) relative to Co_{5.47}N, which meant that the lattice expansion was due to the partial substitution of Co by Fe ⁵⁵. With the increase of pyrolysis temperature, the diffraction peak intensity belonging to the (002) crystal plane of graphitic carbon increased, the graphitization degree gradually increased as well as the electronic conductivity. With the increase of the urea dosages during the in situ pyrolysis process, the diffraction peak intensity of 43.7deg belong to the (111) plane of Co_{5.47}N increased continuously, indicating that the coordination between metal and nitrogen gradually increased, and nitrogen was successfully introduced into the lignin-derived carbon support. The optimal Fe/Co ratio and urea dosage was beneficial to the formation of CoFe alloy and Co_xN heterojunction.



Fig. 1. (a) Synthesis route of CoFe-Co_xN@NC heterojunction, XRD pattern of CoFe-Co_xN@NC with various (b) metal ratios, (c) pyrolysis temperatures and (d) urea dosages

The delicate morphology of CoFe-Co_xN@NC was investigated by SEM, TEM and HRTEM, as shown in Fig. 2. Compared with the metal free sample NC, the pore structure of CoFe-Co_xN@NC exposed more active sites and the metal particles were highly dispersed without agglomeration, thus would improve the catalytic activity. In addition, HRTEM images of CoFe-Co_xN@NC in Fig. 3d-f showed the clear lattice of CoFe alloy nanoparticles and the lattice spacing was 0.202 nm, which can be ascribed to the (110) diffraction plane of CoFe alloy; a distinct lattice fringe of 0.207 nm, consistent with the (111) lattice plane of Co_{5.47}N; the lattice spacing of the outer carbon layer is about 0.34 nm, attributing to the *d* -spacing of the (002) plane in graphitic carbon. Moreover, the energy dispersive X-ray spectrometer (EDX) showed that Co, Fe, C, N and O elements were uniformly distributed along the surface of CoFe-Co_xN@NC (Fig. 2g). CoFe-Co_xN heterojunction particles with sizes ranging from 20.2 nm to 58.6 nm were encapsulated by the graphitic carbon layer and uniformly distributed on the carbon support. The SEM-EDX image of CoFe-Co_{5.47}N@NC

was shown in Fig. S5a, and the average particle size was 31.36 nm and the particle size distribution was shown in Fig. S5b. The results above proved the formation of CoFe alloy and Co_xN heterojunction encapsulated by the graphic carbon layers and located on the nitrogen-doped lignin-derived shell-core carbon support.



Fig. 2. SEM image of (a) NC and (b-c) CoFe-Co_xN@NC, (d) TEM image, (e-f) HRTEM image and (g) elemental mappings of Co, Fe, C, N and O of CoFe-Co_xN@NC

Raman spectroscopy was conducted to study the effect of pyrolysis temperatures on the composition of CoFe-Co_xN@NC, as shown in Fig. S6a. The D-band at 1340 cm⁻¹ and the G-band at 1590 cm⁻¹ represented the defect degree and graphitic degree of carbon, respectively. The degree of defects and graphitization could be measured by the intensity ratio of I_D/I_G . As the pyrolysis temperature increased from 500 to 800°C, the I_D/I_G increased from 0.92 to 1.12; when further increased the pyrolysis temperature, the I_D/I_G decreased. The high degree of defect structure would lead to the increase of the adsorption capacity while the high degree of graphite structure would lead to the increase of the adsorption capacity while the high degree of graphite structure would lead to the increase of electrical conductivity of the carbon material. The results suggested that CoFe-Co_xN@NC pyrolysis at 700°C had the most suitable degree of defects and graphitization, thus had relative high adsorption capacity and high electrical conductivity, which was beneficial to promote the OER electrocatalytic performance. The porous character of CoFe-Co_xN@NC was investigated by the N₂ adsorption/desorption curve, indicating that CoFe-Co_xN@NC had the characteristics of hierarchical porous structure. Its specific surface area was 195.9 m²·g⁻¹ and the pore volume was 0.27 cm^{3*}g⁻¹. The macro- and mesoporous structure of the carbon material facilitated the adsorption transfer of OH⁻ reactants and O₂ products in OER, and the large specific surface area facilitated the contact of the active sites.

The chemical composition and valence state of CoFe-Co_xN@NC were determined by XPS in Fig. 3 and Fig. S8. The XPS survey spectrum revealed the presence of Co, Fe, O, N and C elements, which was consistent with the results of elemental mappings by TEM-EDX. The C 1 _S spectrum confirmed the existence of C-C (284.8 eV), C-N (285.4 eV), C-O (286.6 eV), C=O (288.7 eV) and O-C=O (290.8 eV) in CoFe-Co_xN@NC⁵⁶, C-N bond indicated that nitrogen was successfully doped into the carbon skeleton, while the existence of C=O and O-C=O implied substantial oxygen-containing groups decorated on the surface which can create defects in the carbon matrix. The N 1 s spectrum at the peaks of 398.6, 400.1, 401.1 and 403 eV corresponded to pyridine N (38.53%), pyrrole N (17.99%), graphite N (33.45%) and pyridine N oxide (10.04%), respectively 57,58 , the abundant pyridinic N can coordinate with metal atoms through metal–N–C structure to optimize

the local electronic structure, graphitic N was expected to improve the diffusion-limiting current, both benefiting the electrochemical process. In addition, the surface element content of CoFe-Co_xN@NC with various urea dosages and the relative contents of the four nitrogen atoms were listed in Tab. S1 and S2, respectively. The hydrophilicity of carbon materials could be improved by the surface functional groups to balance the adsorption and desorption of intermediates and products during OER performance. As CoFe alloy in CoFe-Co_{5.47}N@NC catalyst was encapsulated by the graphitic carbon layer, the spectral intensity of Fe 2p and Co 2p spectra in CoFe-Co_xN@NC catalyst was relatively weak, as shown in Fig. 3c-d. The Fe 2p spectrum of Fe@NC and the Co 2p spectrum of Co@NC was shown in Fig. S8. For the Fe 2p spectrum, the first double peak was 711.5 and 724.2 eV, and the second double peak was 713.9 and 727.3 eV, designated as Fe²⁺ and Fe³⁺, respectively^{59,60}, and the peaks near 719.1 eV and 735.2 eV were classified as the satellite peaks of Fe 2p^{3/2} and Fe 2p^{1/2}, respectively. Similarly, Co 2p spectrum showed the zero-valence state (778.3 and 795.2 eV) and the ionic state (780.0 and 797.4 eV) Co with satellite peaks (785.6 and 803.4eV), which are derived from CoFe alloy and Co_xN species, respectively.

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Fig. 3. High-resolution XPS spectrum of CoFe-Co_xN@NC: (a) C 1s, (b) N 1s, (c) Fe 2p, (d) Co 2p

3.2 Electrocatalytic performance of CoFe-Co_xN@NC

The OER electrocatalytic activity of CoFe-Co_xN@NC catalysts as investigated in a standard three-electrode system in 1.0 M KOH solution. The effects of the pyrolysis temperature, urea dosage, Fe/Co metal ratio and metal/lignin ratio during synthesis process on the OER electrocatalytic activity of CoFe-Co_xN@NC were shown in Fig. 4 and Fig. S9-10. As the pyrolysis temperature increased, the overpotential required to achieve a current density of 10 mA*cm⁻² (η 10) using CoFe-Co_xN@NC catalyst decreased at first and then increased at the temperature of 700, which was related to the degree of defects and graphitization of the carbon material. The low pyrolysis temperature leaded to incomplete carbonization of the sample, thus showed poor electrical conductivity that seriously affected the OER electrocatalytic activity. Excessive pyrolysis temperature leaded to the metal agglomeration and pore blockage, thus reduced the OER electrocatalytic activity. When the optimal ratio of urea and EHL-COOH-CoFe precursor was 1:1, CoFe-Co_xN@NC exhibited the lowest overpotential $\eta 10$ of 270 mV, that was much lower than that of CoFe@C catalyst without the doping of urea (340 mV). The active sites of pyridine N and graphite N could promote the synergistic interaction between nitrogen and CoFe alloy, the Co_xN composition in CoFe-Co_xN@NC would increase from the XRD patterns (Fig. 1d) with the increasing dosage of urea during the in situ pyrolysis process, therefore, the composition ratio of CoFe alloy and Co_xN heterojunction decreased that caused the inferior OER electrocatalytic activity of CoFe-Co_xN@NC.

Monolithic Co-based catalyst was insufficient to deliver the ideal OER electrocatalytic activity and stability. Rational synthesis of bimetallic CoFe alloy was investigated through the partial substitution with Fe, and the metal composition content in CoFe-Co_xN@NC determined by ICP-OES was shown in Table S3. With the increase of substitution with Fe based on the atomic ratio of Fe and Co to 21%, the overpotential η 10 was decreased to 270 mV, while that of Co@NC was 340 mV; further increased the Fe composition content, Fe became the major active composition, Co_xN content in CoFe-Co_xN@NC decreased and ever disappeared verified by the XRD patterns in Fig. 1b, thus decreased the overpotential η 10 to 416 mv with the Fe substitution of 96%, while that of Fe@NC was 448 mV. With the increase of the ratio of metal moles and lignin to 8 mM/g in the self-assembly coordination with Fe³⁺ and Co²⁺ process, CoFe-Co_xN@NC showed the optimal OER performance. Based the results above, the optimal synthesis parameters were the pyrolysis temperature of 700, the ratio of metal moles and lignin of 8 mM/g, the atomic ratio of Fe:Co = 2:6, the ratio of urea and EHL-COOH-CoFe precursor of 1:1, and the metal content of the corresponding CoFe-Co_xN@NC was 1.45 wt% of Fe and 5.60 wt% of Co. The overpotential $\eta 10$ of CoFe-Co_xN@NC material was 270 mV with a Tafel slope of 85 mV·dec⁻¹, which could be comparable to the performance of commercial Ir/C catalyst (252 mV@10 mA*cm⁻², 82 mV*dec⁻¹).

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Fig. 4. Effect of (a-b) pyrolysis temperature, (c-d) urea dosage and (e-f) Fe/Co ratio on the OER electrocatalytic activity of CoFe-Co_xN@NC by LSV measurement and its corresponding Tafel slop

The stability of electrocatalyst was another key feature in practical applications. The stability of CoFe-Co_xN@NC for OER was evaluated by the chronopotentiometri measurement recorded at a constant current density of 10 mA*cm⁻², as show in Fig. 5. After the continuous test for 80000 s, the overpotential of CoFe-Co_xN@NC kept well-maintained, of which just increased by about 20 mV. The TEM images, TEM-mapping images and XRD pattern (Fig. S11-12) of CoFe-Co_{5.47}N@NC catalyst after stability test indicated that the structure and morphology of the catalyst remains unchanged. From the electrocatalytic performance above, the catalytic activity and stability of CoFe-Co_xN@NC for OER was mainly attributed to the active sites of CoFe-Co_{5.47}N heterogeneous structure encapsulated by the graphic carbon layers and embedded in the nitrogen-doped lignin-derived shell-core carbon material.

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Fig.5. (a) the LSV curve of CoFe-Co_xN@NC for OER before and after the 80000 s cycle at a constant current density of 10 mA*cm⁻², (b) the chronopotentiometri measurement of recorded at a constant current density of 10 mA*cm⁻² for CoFe-Co_xN@NC

3.3 Electrocatalytic mechanism

In order to provide insights into the electrocatalytic mechanism of CoFe-Co_xN@NC for OER, DFT calculation was employed to investigate and compare the OER catalytic performances of four catalyst models including naked FeCo alloy, FeCo alloy encapsulated by carbon layer (CoFe@C), CoFe alloy encapsulated by nitrogendoped carbon layer (CoFe@NC), and the as-prepared CoFe-Co_xN@NC. Generally, OER undergoes a fourelectron step pathway in a basic electrolyte showed as Fig. 6a. The Gibbs free energy profiles for four catalysts were depicted in Fig. 6b, and the rate-determining step (RDS) with the value of the Gibbs free energy barrier was highlighted, where a fairly high Gibbs free energy barrier of 1.71 eV was obtained at 1.23 V for OER over naked CoFe alloy. The reason for this mainly attributed to the strong absorption (a Gibbs free energy barrier of -1.93 eV) of the active oxygen on the surface of CoFe alloy to form an excessively stable *O intermediate for OER. The charge density differences presented in Fig. S13 showed that charges were almost transferred onto the surface of oxygen atom, further revealing the enhanced electrostatic interaction between oxygen and CoFe alloy. In such case, more energies were required to overcome the barrier from *O to *OOH, resulting in the high overpotential for OER over CoFe alloy. Theoretically, CoFe alloy encapsulated by graphitic carbon layer can prevent it from the direct contact with reactive oxygen species, however, the interfacial distance between CoFe alloy and graphitic carbon layer would affect the stability and charge transfer. It can be seen from Fig. 6c that the total energy of the CoFe@C heterojunction catalyst was lowest when the interfacial distance reached 4A. Nevertheless, the electrons mainly accumulated in its interfacial space at its most stable configuration based on the charge density difference showed in Fig. 6c inserted images, making the active site to *OH pathway as the RDS with the high energy barrier of 1.56 eV. Interestingly, doping nitrogen in the graphitic carbon layer can redistribute the charge density without sacrificing its stability. The electrons transferred from CoFe alloy can be well distributed on the nitrogen-doped graphitic carbon layer as described in Fig. 6d-e, promoting the OER process with a lower energy barrier of 0.68 eV at RDS from *O to *OOH. Actually, in this study, in situ nitrogen-doping pyrolysis process can form a specific heterojunction CoFe-Co_xN@NC with three phases, nitrogen-doped Co_{5.47}N-connected CoFe alloy. The Co_xN as a linker can induce the uniform distribution of charge density as shown in Fig. 6f based on the DFT simulation, which could optimize the adsorption/desorption strength of the OER intermediates and products. As a result, a lowest limiting energy barrier of 0.43 eV can be achieved by DFT calculation at 1.23 V for OER over CoFe-Co_xN@NC (Fig. 6b) in comparison with other used catalysts, in consistent with the trend of experimental results.



Fig. 6. Catalytic activity evaluated by DFT simulation. (a) Proposed reaction circle with geometryoptimized intermediates for OER over CoFe alloy and CoFe-Co_xN@NC, respectively. (b) Gibbs free energy profiles at 1.23 V for OER over CoFe alloy, CoFe@C, CoFe@NC and CoFe-Co_xN@NC, respectively. (c)

Effect of interfacial space between CoFe alloy and carbon layers on the stability of heterojunction catalysts, where yellow and cyan regions indicate increased and decreased charge distributions, respectively in the inserted images. (d) 2D and (e) 3D display of charge density difference of CoFe@C doped with N element (CoFe@NC). (f) Charge density difference of intermediates for OER over CoFe-Co_xN@NC

4. Conclusions

In situ coupling strategy of CoFe alloy and $Co_x N$ heterojunction encapsulated and embedded in the nitrogendoped lignin-derived carbon CoFe-Co_xN@NC was provided to create a novelty highly active and stable electrocatalyst for oxygen evolution reaction. Biorefinery lignin was first carboxylated to improve its solubility and precisely coordinated with metal ions to form the coordination polymer precursor, that promoted the dispersion of metal active component and the encapsulated grapheme lawyer of CoFe alloy and $Co_x N$ heterojunction. CoFe-Co_xN@NC exhibited an excellent OER activity with the lowest overpotential of 270 mV at 10 mA·cm⁻² and stability with the only increment of 20 mV after 80000s in the alkaline media, comparable to that of commercial Ir/C catalyst. The as-prepared catalyst with high activity and cycle stability was attributed to the synergistic effect of CoFe alloy, $Co_x N$ and nitrogen-doped encapsulated grapheme lawyer, that reduce the adsorption strength between *O and CoFe alloy, prevent metals leaching and agglomeration, and improve the electron transfer efficiency. This method not only turned waste into treasure from biomass waste lignin, but also provided a solution for the replacement of precious metal catalysts in the field of renewable energy conversion and storage.

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Graphical Abstract